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1 2	On the morphology of cellulose nanofibrils obtained by TEMPO-mediated
3	oxidation and mechanical treatment
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18 19 20 21	Abstract
22 23	The morphological properties of cellulose nanofibrils (CNF) obtained from eucalyptus
24	pulp fibres were assessed. Two samples were produced with the same chemical treatment
25	(NaClO/NaBr/TEMPO (2,2,6,6-tetramethylpiperidine-1-oxyl radical) oxidation), but distinct
26	mechanical treatment intensities during homogenization. It was shown that the nanofibrils
27	production yield increases with the mechanical energy. The effect of mechanical treatment on
28	the yield was confirmed by laser profilometry of air-dried nanocellulose films. However no
29	significant differences were detected regarding the nanofibrils width as measured by atomic
30	force microscopy (AFM) of air-dried films. On the other hand, differences in size were found
31	either by laser diffraction spectroscopy or by dynamic light scattering (DLS) of the cellulose
32	nanofibrils suspensions as a consequence of the differences in the length distribution of both
33	samples. The nanofibrils length of the more nanofibrillated sample was calculated based on
34	the width measured by AFM and the hydrodynamic diameter obtained by DLS. A length

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35	value of ca. 600 nm was estimated. The DLS hydrodynamic diameter, as an equivalent
36	spherical diameter, was used to estimate the CNF length assuming a cylinder with the same
37	volume and with the diameter (width) assessed by AFM. A simple method is thus proposed to
38	evaluate the cellulose nanofibrils length combining microscopy and light scattering methods.
39	
40	Keywords: Nanofribrillated cellulose; TEMPO-oxidised cellulose nanofibrils; morphology;
41	size; dynamic light scattering, atomic force microscopy
42	
43	Introduction

In the last decade the interest in cellulose nanofibrils (CNF) increased exponentially. This 44 45 material, usually obtained as a viscous gel, was firstly produced in the eighties (Turbak et al., 46 1983) after passing a wood fibre suspension several times through a homogenizer under high pressure. Chemically pre-treated CNF can be regarded as interconnected webs of tiny 47 48 nanofibrils with diameters (D) typically of less than 20 nm and lengths (L) in the micrometer 49 scale. The specific D and L dimensions of CNF can vary according to the pristine material 50 and the conditions of the process used for their preparation but the aspect ratio (L/D) is in all 51 cases very high (Gardner et al., 2008; Lavoine et al., 2012; Shinoda et al., 2012). CNF 52 possesses a specific surface area significantly higher than that of the pristine cellulose fibres (values can reach more than $100 \text{ m}^2/\text{g}$ (Lavoine et al., 2012)). However, the exact values have 53 54 been difficult to assess. Nitrogen adsorption using BET calculation has been the most 55 common method to determine the specific surface area but it is widely accepted that the BET 56 values underestimate the real values of surface area, because the measurement is made on the 57 material after drying, where the microfibrils are strongly aggregated by hydrogen bonding 58 (Lavoine et al. 2012). In addition, the BET specific surface area will be also strongly 59 dependent on the applied drying process (air drying, freeze drying, spray drying). One of the

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60 most important characteristics of CNF is the high strength property, commonly measured on 61 CNF films, that is the result of a high specific surface area for hydrogen bonding. Besides, 62 highly translucent films can be obtained from the CNF suspensions. All these characteristics 63 of CNF, namely the viscosity properties of the CNF gel, the high specific surface area, high 64 mechanical resistance and high light transmittance of its solid films make it appealing for a 65 wide range of applications. Just to mention a few, it may be used in formulations as a 66 viscosity modifier, as gel for biomedical applications (Chinga-Carrasco and Syverud, 2014), 67 as a mechanical reinforcement material in composites, including paper (Ahola et al., 2008a; 68 Syverud and Stenius, 2009), for paper coating (Brodin et al., 2014), in films for food 69 packaging (Syverud and Stenius, 2009; Aulin et al., 2010) and for electronic devices (Chinga-70 Carrasco et al., 2012), and as gas barrier material (Fukuzumi et al., 2009, Lavoine et al. 2012). 71 The assessment of the physical and chemical properties of cellulose nanofibrils produced 72 by several methods is of main interest from both the fundamental and practical point of view. 73 Accordingly to a previous review, the main points that should be addressed are the amount of 74 produced nanomaterial, the rheology of the dispersion, the particle size (including aspect 75 ratio) and size distribution, crystallinity, specific surface area, surface charge and chemistry, 76 and mechanical properties (Kangas et al., 2014). Obviously, in account for the foreseen 77 applications, some of the properties can have more importance than the others.

The size and size distribution of nanofibrils is always an important parameter to consider, but it should be mentioned that presently no standard methods or validated techniques are available for the size evaluation of polydisperse nanomaterials with a high aspect ratio (Fraschini et al., 2014). Microscopy techniques, such as high-resolution scanning electron microscopy (SEM), transmission electron microscopy (TEM) and atomic force microscopy (AFM) are most appropriate to visualize the cellulose nanofibrils (Chinga-Carrasco et al., 2014). However, it is widely recognised that these are much dependent on the operator and

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85 are usually restricted to the analysis of a small amount of sample, besides being time-86 consuming (Fraschini et al., 2014). On the other hand, techniques based on light scattering, 87 such as dynamic light scattering (DLS), can circumvent most of the drawbacks mentioned for 88 microscopy-based techniques provided that the particles are spheres or have a shape close to 89 that of spheres. This is not the case of cellulose nanofibrils that are closer to a cylindrical 90 shape. Notwithstanding, it was reported for cellulose nanocrystals that the equivalent 91 hydrodynamic radius, measured by DLS, did not differ much from the theoretical 92 hydrodynamic radius, calculated for cylinder-shaped particles based on the dimensions of 93 length and width assessed by Field emission-SEM (Fraschini et al. 2014). Thus, microscopy 94 and light scattering methods are considered complementary.

95 In the present work, cellulose nanofibrils obtained by TEMPO-mediated oxidation and 96 different intensity of mechanical treatment will be assessed using complementary techniques, 97 covering structures from the micrometre to the nanoscale. Special emphasis will be put on the 98 determination of the cellulose nanofibrils length using DLS and AFM data and different 99 computing approaches.

100

101 Experimental section

102 Nanofibrils preparation

A bleached eucalyptus kraft pulp was pre-treated with NaClO and catalytic amounts of TEMPO (2,2,6,6-tetramethylpiperidine-1-oxyl radical) and NaBr according to a methodology described elsewhere (Saito et al. 2007). In a typical experiment, 15 g of cellulose fibres were dispersed in distilled water containing TEMPO (0.016 g per g of fibres) and NaBr (0.1 g per g of fibres) at a consistency of 1%. The mixture was stirred during 15 minutes at room temperature in order to assure a good dispersion of all the substances. After this, a 15% sodium hypochlorite solution was added drop by drop to the slurry. The volume of NaClO

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110 was calculated to have 4 mmol per gram of cellulose. The pH was kept at 10 by addition of 111 drops of a 0.5 M NaOH solution. The oxidation was considered finished when the pH kept 112 constant at 10. The oxidized fibres were then filtered and washed with distilled water. The 113 fibres were then passed through a homogenizer (GEA Niro Soavi Panda Plus 2000, Italy) 5 114 times at 300 bar (CNF-5p) or 15 times (five passes at 300 bar and 10 passes at 600 bar) (CNF-115 15p). Homogenization was performed at room temperature with a pulp consistency of 1.5 %. 116 During the homogenization it is normal that after several passes the temperature of the fluid 117 raises up to 60-70°C. When this happened, homogenization was stopped in order to avoid 118 pump cavitation and the process continued after cooling of the fluid and equipment at room 119 temperature. The resulting nanofibrils were then characterized for their morphology 120 properties, as described below. 121 122 Characterization methods 123 124 125 The yield of nanofibrils production was determined in triplicate after centrifugation (Alila 126 et al. 2013) of 40 mL of CNF suspensions (0.2%, w/w) at 9000 rpm for 30 min: the retained 127 fraction was analysed for its solid content and compared to the original to obtain by difference 128 the percentage (w/w) of supernatant material. The percentage of fibrillar material separated at 129 the supernatant by centrifugation, corresponds thus to the "yield". The transmittance of CNF 130 suspensions (0.1%, w/w) in the 400-800 nm visible range was measured using a Jascow V550 131 spectrophotometer. Transmittance was measured immediately after stirring the CNF 132 suspensions. 133 Field emission-SEM, AFM micrographs and laser profilometry (LP) were taken on CNF

films. The films were obtained by air drying of the original nanocellulose suspensions (0.2%),

135 w/w) in a Petri plate at room temperature for about 7 days.

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The Field emission-SEM images were acquired at 1000x magnification with a Zeiss Ultra field-emission SEM, using 3 kV and 4.7 mm acceleration voltage and working distance, respectively.

AFM imaging was performed using a Multimode AFM microscope equipped with a Nanoscope V controller (Digital Instruments). All images were recorded in ScanAsyst mode (peak force tapping mode), at room temperature, in air. The AFM tips, of spring constant value ~0.4 N/m, were purchased from Bruker AFM probes. The size of the assessed areas was $2 \mu m \times 2 \mu m$. The nanofibril diameter distribution was quantified on the AFM images, as previously described by Chinga-Carrasco et al. (2011).

145 For LP analysis, samples of 10 mm \times 10 mm were coated with a layer of gold (Lehmann, 146 Lehman Mess-Systeme AG, Baden-Dättwil, Germany). Ten topography images were 147 acquired from the top and bottom sides (bottom refers to the part of the film that during the 148 film formation has been in physical contact with the Petri plate and top refers to the side of 149 the film formed in contact with air). The lateral and z-resolution of the profilometry system was 1 μ m and 10 nm, respectively. The size of the local areas was 1 mm \times 1 mm. The 150 151 surfaces were horizontally levelled. The surface images were bandpass filtered to suppress the 152 surface structures with wavelengths larger than approximately 160 µm, applying a FFT filter 153 implemented in the ImageJ program. The roughness described by the root-mean square (Sq) 154 was thus quantified at wavelengths of less than 160 µm (Chinga-Carrasco et al. 2014).

Laser diffraction spectroscopy (LDS), which is appropriate to analyze particles with size larger than 1 μ m, according to the Lorentz-Mie theory (Gouesbet and Grehan, 1999), was performed using a Mastersizer 3000 (Malvern Instruments). 100 mL of the original CNF suspensions with a dry matter-content of approximately 0.1% were prepared and magnetically stirred during one hour before the measurements. The suspensions were analysed prior to centrifugation, thus including all the material obtained after the chemi-mechanical treatment

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of the pulp fibres. A certain volume of the prepared suspension was added to 700 mL of water in the equipment vessel until a 10–20% obscuration was reached, and the tests were performed setting the pump speed to 2000 rpm. The results presented are an average of six measurements.

165 Dynamic light scattering (DLS) measurements were made using a Zetasizer Nano ZS 166 (Malvern Instruments). This technique allows the analysis of particles in the size range 167 between 0.6 nm and 6.0 μ m. The detection was made at a scattering angle of 173° and the 168 intensity size distributions were obtained from analysis of the correlation function using the 169 CONTIN algorithm in the instrument software. The supernatants obtained after centrifugation 170 of 40 ml of the CNF suspensions (0.2% w/w) at 9000 rpm for 30 min were analysed, and the 171 results were an average of five replicated measurements. Zeta potential measurements were 172 carried out in triplicate using the same equipment. 173 A systematization of the part of the analysed CNF sample, the evaluated property and the

- technique used for that is presented in Table 1.
- 175
- Table 1. The properties of CNF assessed, the different used techniques, and the part of the
 sample considered for analysis

Part of the analysed sample	Property	Technique
	Light transmittance	Spectrophotometry
		Field emission-SEM
All	Surface morphology	Atomic Force Microscopy
		Laser profilometry
	Particle size	Laser diffraction spectroscopy
V	Yield	Gravimetric analysis
Supernatant	Particle size	Dynamic light scattering
	Particle charge	Zeta potential measurement

178

- 179 **Results and discussion**
- 180

181 General characterization of CNF

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183	Two samples of cellulose nanofibrils were produced using the same TEMPO-mediated
184	oxidation pre-treatment but different extents of the subsequent mechanical treatment in a
185	homogenizer. Before the evaluation of nanofibrils size and surface properties, the suspensions
186	were characterized for the yield of nanofibrils production, zeta potential and transmittance
187	(Table 2). The zeta potential values of the two samples (-41 and -46 mV, for CNF-5p and
188	CNF-15p, respectively) are both negative and of similar magnitude. The negative values are
189	mostly due to the presence of carboxylates (COO ⁻) at the surface of nanofibrils generated
190	during the oxidative pre-treatment with NaClO/NaBr/TEMPO.
191	As expected, the yield of the nanofibrils sample obtained after 15 passes (95%) was much
192	higher than that obtained after 5 passes (63%), in agreement with the higher intensity of the
193	mechanical treatment. The visible spectra in the transmittance mode (Fig. 1) also evidenced
194	higher transmittance for CNF-15p, corresponding to a clearer suspension and corroborating

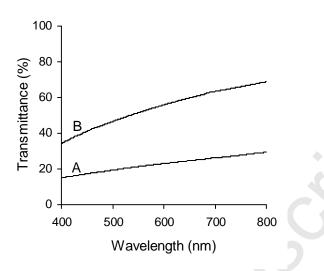
195 the higher amount of nanosized material in this dispersion.

196

197 Table 2. Results on the production of cellulose nanofibrils by TEMPO-mediated198 oxidation

Sample	Yield (%)	Zeta Potential (mV)	Transmittance (600 nm, %)
CNF-5p	63±3	-41±4	23
CNF-15p	95±1	-46±3	56

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Fig. 1. Visible spectra in the transmittance mode of 0.1% suspensions of CNF-5p (A) and

CNF-15p (B)

201

202

203 Surface morphology of CNF films

204 Field emission-SEM provided an insight into the surface morphology of the CNF films 205 and, indirectly, confirmed the relative amount of nanofibrillated material. Based on the Field 206 emission-SEM images (Fig. 2) one can observe that the CNF-5p has a major fraction of 207 residual fibres. The amount of residual fibres was reduced in the CNF-15p sample due to the 208 additional homogenization steps. These observations were corroborated by the LP analysis. 209 The LP surface roughness (Rq) assessed at various wavelengths was significantly lower for 210 the films containing higher amount of nanofibrils (CNF-15p) (Fig. 3). The LP-roughness of 211 the CNF-15p sample confirms that this material is highly fibrillated and contains a major 212 fraction of cellulose nanofibrils (Chinga-Carrasco et al., 2014; Chinga-Carrasco and Syverud, 213 2014). The higher homogeneity of the suspension of CNF-15p (transmittance of 56% at 600 214 nm, Table 2) and the corresponding higher amount of nanofibrils (yield of 95% of nanofibril 215 production) produce relatively smooth film surfaces (Fig. 2 and 3).

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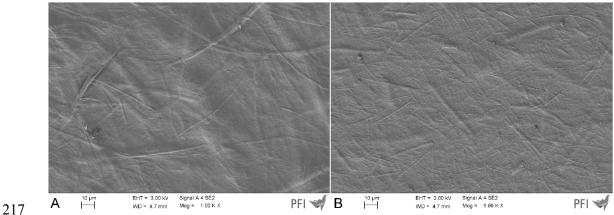
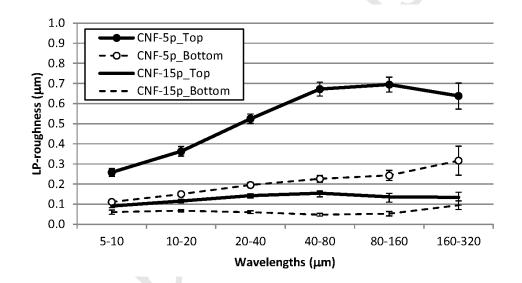




Fig. 2. Field emission-SEM of films of CNF-5p (A), and CNF-15p (B).



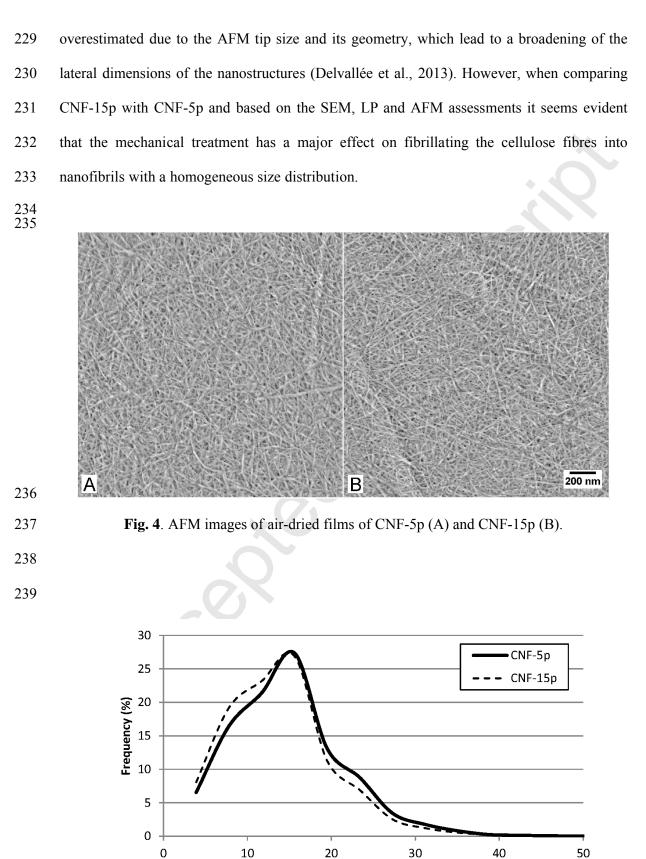
219

Fig. 3. Laser profilometry topography assessment (top and bottom refer to the sides of the
film formed in contact with air and Petri plate, respectively)

222

As mentioned, AFM is a valuable technique that may be used for the assessment of the surface roughness at the nanoscale and of the cellulose nanofibrils morphology. In this work, the nanofibrils width was estimated based on AFM images of nanocellulose films (Fig. 4 and Fig. 5). A mean value of approximately 15 nm was quantified for both samples (the average width based on 4 images from each sample was 15 nm \pm 1.6 and 14 nm \pm 0.7 for CNF-5p and CNF-15p, respectively). Keep in mind that the mean value of the nanofibril widths may be

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Nanofibril diameter (nm)

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24	1
_	1

Fig. 5. Nanofibril diameter distribution obtained by AFM for samples CNF-5p and CNF-

243

244

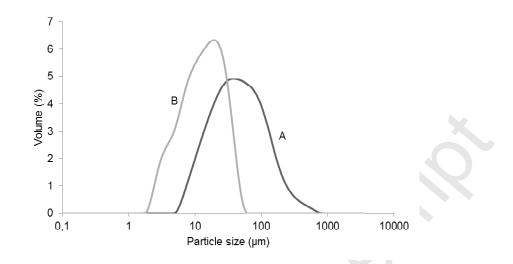
15p.

245 Complementary optical methods for the assessment of nanofibril biometry

246 Microscopy based techniques such as AFM but also SEM and TEM are adequate to assess 247 the morphology of cellulose nanofibrils and hence they have been widely used (e.g. Saito et 248 al., 2006; Ahola et al., 2008b; Pääkkö et al., 2007; Abe et al., 2007; Fukuzumi et al., 2009; 249 Chinga-Carrasco et al., 2011). However, as mentioned in the introduction section, they may 250 be time-consuming (sample preparation, image acquisition and analysis), operator dependent 251 and restricted to the analysis of a small amount of sample. These drawbacks are particularly 252 relevant for the measurement of the nanofibrils length, which presents usually a broad 253 distribution and, besides, cannot be evaluated in images like those of Fig. 4. Therefore, in this 254 study, an attempt was made to use relatively simple and fast methods to assess the length of 255 the cellulose nanofibrils. For that, laser diffraction spectroscopy (LDS) and particularly 256 dynamic light scattering (DLS) measurements were employed. The latter has been used for 257 nanofibrils of distinct origins, as detailed in the open literature (Mandal and Chakrabarty, 258 2011; Qua et al, 2011; Beck et al. 2012; Zhou et al, 2012; Morais et al. 2013; Boluk and 259 Danumah 2014; Fraschini et al. 2014).

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Fig. 6. Volume distributions obtained by laser diffraction spectroscopy for samples CNF-5p

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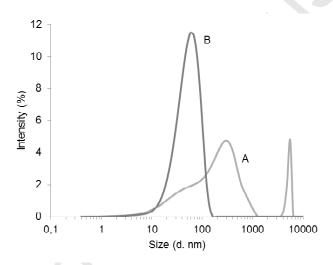
(A) and CNF-15p (B).

264

265 As for the laser diffraction spectroscopy, it is more convenient to particles in the 266 micrometer range, which is not really the case of the cellulose nanofibrils in this study. 267 However, taking into account that the produced nanofibrils suspensions also show some non-268 completely fibrillated content (as suggested by the yield of the production process), this 269 technique was employed to detect and analyse the fraction of the larger material as well as of 270 the nanofibrils aggregates (> $1 \mu m$). The particle size distributions obtained by LDS for CNF-271 5p and CNF-15p are shown in Fig. 6. It should be noted that these are volume (and not 272 number) distribution curves and therefore even the presence of a small number of less-273 fibrillated material and nanofibrils aggregates has a great impact on the size distribution. As 274 clearly shown in Fig. 6, the volume distribution of the CNF-5p sample is shifted to higher size 275 values. This may be attributed to the higher amount of less-fibrillated material presented in CNF-5p (see Table 2, Fig. 2 and Fig. 3). Moreover, it is important to note that LDS provides 276 277 an equivalent diameter based distribution assuming that the light scattering pattern of the 278 material is identical to that of spherical particles. For micro- and nano-fibrils with a high 279 aspect ratio, this approach leads to values that can be used only for the comparison between

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280 samples and not as a direct measure of the real size of the material. With this in mind, the size 281 reduction from CNF-5p to CNF-15p (Fig. 6) is obvious, when these samples are analysed as a 282 whole (i.e., without any further step of fraction selection such as centrifugation). 283 Considering the existence of nanofibrillated material, dynamic light scattering was used 284 since it is more appropriate than LDS to assess particle sizes in the nanoscale range. To get 285 more information on the produced nanofibrils size, suspensions of CNF-5p and CNF-15p 286 were centrifuged (in the same way as that used to determine the nanofibril yield) and only the 287 supernatants were analysed. The size distributions of the supernatants are depicted in Fig. 7.



288

Fig. 7. Intensity distributions obtained by dynamic light scattering for supernatants obtained
after centrifugation of CNF-5p (A) and CNF-15p (B).

291

While sample CNF-15p presents an unimodal distribution with a mode at *ca*. 55 nm, the sample CNF-5p presents a broad range of sizes from about 5 nm to 1000 nm and some objects around 5000 nm. The DLS distribution of CNF-15p is more in agreement to what is expected for a highly nanofibrillated material (Mandal and Chakrabarty, 2011; Zhou et al, 2012; Fraschini et al. 2014). From these results, it is evident that the supernatant of sample CNF-15p presents a greater uniformity in size distribution comprising only nanofibrils. The supernatant

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298 of CNF-5p, on the other hand, still denotes some residual fibres (or nanofibrils aggregates) of 299 higher dimension, in spite of the preliminary centrifugation process.

300 Due to the lack of homogeneity and broad size range observed for CNF-5p, it is not 301 possible to relate the corresponding results obtained by DLS and AFM, contrary to the case of 302 the CNF-15p sample. For the latter, that shows a simple size distribution, several attempts 303 were made to obtain the length (L) of the nanofibrils by combining the DLS and AFM data, 304 *i.e.*, by considering the hydrodynamic diameter (mode) of 56 nm measured on the supernatant 305 by DLS (D_{DLS}) and the average width of 14 nm measured on the films by AFM (W_{AFM}). In a first approach, the Tirado and Garcia de la Torre formula for the ratio $D_{\rm HC}/D_{\rm HS}$, given by Eq. 306 307 1, was considered (Tirado and García de la Torre, 1980; Fraschini et al., 2014). Here, D_{HC} and 308 $D_{\rm HS}$ are the hydrodynamic diameters, respectively, of a cylinder and a sphere, having the 309 same volume; the term γ is defined in Eq. 2. Taking $D_{\rm HC} = D_{\rm DLS}$, $W = W_{\rm AFM}$ and $D_{\rm HS} =$ $(3/2W^2L)^{1/3}$, an average nanofibril length (L) of 150 nm was obtained by solving the Eq. 1. 310 311 However, to apply the formula described by Eq. 1, the aspect ratio of the cylinder should be 312 between 2 and 20 which may not be the present case (Lavoine et al. 2012). Thus, the 313 calculation of the length was also done using a formula developed for a wider range of aspect 314 ratios (up to 100) (Hansen, 2004). The result obtained was the same (L=150 nm).

315

3

16
$$\frac{D_{\rm HC}}{D_{\rm HS}} = \frac{\left(\frac{2}{3}\right)^{1/3} \left(\frac{L}{W}\right)^{2/3}}{\ln\left(\frac{L}{W}\right) + \gamma}$$
 Eq (1)

317
$$\gamma = 0.312 + 0.565 \left(\frac{W}{L}\right) - 0.100 \left(\frac{W}{L}\right)^2$$
 Eq (2)

318 In a second approach, a very simple calculation was carried out considering the mode of 319 the intensity distribution obtained by DLS as an equivalent spherical diameter and using it to

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320 compute the volume of the nanofibrils and deriving the length of a cylinder with the same 321 volume and with a width assessed by AFM, as shown in Equations 3 and 4. Note that the 322 main difference between the two approaches is that in the first approach the hydrodynamic 323 diameter obtained by DLS is taken as a cylinder hydrodynamic diameter $(D_{\rm HC})$ while in the 324 second approach it is considered as an equivalent spherical diameter ($D_{\rm HS}$). Using the second 325 approach a value of 597 nm was obtained for L. This value is consistent with the lengths 326 determined by TEM of TEMPO-oxidised cellulose nanofibrils having similar production yield 327 (>95%) (Fukuzumi et al., 2013) or produced with the same amount of NaClO (4 mmol per 328 gram of cellulose) (Shinoda et al., 2012). For instance, the latter authors obtained a number 329 average length of 658 nm for CNF. As for the other calculated values of L, they seem too low 330 for nanofibrils, unless only the shortest nanofibrils are being evaluated. Note also the 331 nanofibril lengths observed in Fig. 4 (AFM images). Although it is difficult to quantify the 332 length due to the entanglement of the nanofibril network, it can be observed that the lengths 333 are at least >500 nm which gives supportive evidence for our proposed approach. 334 Additionally, as stated above the AFM tip overestimates the nanofibrils width, which implies 335 that the lengths are most probably larger than the value estimated in this study. A more 336 accurate measure of nanofibril width could be undertaken with TEM, which will be explored 337 in a future comparative study.

338

339
$$\frac{\pi}{6}(D_{\text{DLS}})^3 = \frac{\pi}{4}(W_{\text{AFM}})^2 L$$
 Eq (3)

340
$$L = \frac{2}{3} \frac{(D_{\text{DLS}})^3}{(W_{\text{AFM}})^2} \qquad \text{Eq (4)}$$

In summary, although different methods were used, some uncertainty remains regarding the real length of the produced nanofibrils. Note that the real length is difficult to assess by dynamic light scattering much due to the high aspect ratio of CNF. Notwithstanding, dynamic

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344 light scattering of the supernatants of nanofibrils suspensions can be considered as a valuable 345 tool to compare the nanofibrils size in terms of relative size trends among a series of 346 preparations. In the case of high yields of produced nanofibrils, the values measured may 347 provide an approximate estimate of nanofibril lengths.

- 348
- 349

350 **Conclusions**

351 Several methods were applied for the analysis and differentiation of CNF obtained by 352 NaClO/NaBr/TEMPO pre-oxidation and mechanical treatment with 5 and 15 passes in a 353 homogenizer, namely CNF-5p and CNF-15p. CNF-15p when compared to CNF-5p showed 354 higher nanofibrils yield and correspondingly higher transmittance in the visible range, while 355 zeta potential was similar due to the same applied chemical oxidation pre-treatment. The 356 width of produced nanofibrils, as assessed by AFM on air-dried films was also similar. On the other hand, surface roughness of the air-dried CNF films increased for the less nanofibrillated 357 358 cellulose.

Techniques different from the conventional microscopic ones to assess nanofibrils size were attempted. Laser diffraction spectroscopy, which is more appropriate to evaluate particles in the micrometer range, showed that the CNF suspension with lower fibrillated content had a volume distribution shifted to higher size value. From the size distributions measured by dynamic light scattering of the supernatants obtained from centrifugation of CNF it was evident that the supernatant of sample CNF-15p presents a greater uniformity in size distribution and only shows nanofibrils.

366 Using the cellulose nanofibrils width measured on the air-dried CNF films by AFM and 367 the hydrodynamic diameter measured on the supernatant by DLS, it was possible to estimate

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368	the nanofibrils length for CNF-15p. Combining microscopy and light scattering data, the
369	average length was thus calculated to be of ca. 600 nm.

370

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- 376

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- 458 Highlights:
- 459
- 460 AFM and dynamic light scattering (DLS) were used for the cellulose nanofibril analysis
- 461 Laser diffraction spectroscopy was used to assess the less-fibrillated material
- 462 The length of cellulose nanofibril could be estimated based on AFM and DLS data
- 463 A value of ca. 600 nm was estimated for the TEMPO-oxidised cellulose nanofibril length
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