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Author: P. Alves R. Cardoso T.R. Correia B.P. Antunes I.J. Correia P. Ferreira



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1 **Surface modification of polyurethane films by plasma and ultraviolet**
2 **light to improve haemocompatibility for artificial heart valves**

3 P. Alves^{a*}, R. Cardoso^a, T.R. Correia^b, B.P. Antunes^b, I.J. Correia^b, P. Ferreira^{a,c}

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5 ^a *CIEPQPF, Departamento de Engenharia Química, Universidade de Coimbra, Polo*
6 *II, Pinhal de Marrocos, 3030-790 Coimbra, Portugal*

7 ^b *CICS-UBI, Centro de Investigação em Ciências da Saúde, Faculdade de Ciências*
8 *da Saúde, Universidade da Beira Interior, 6200-506 Covilhã, Portugal*

9 ^c *IBB, Instituto de Biotecnologia e Bioengenharia, Instituto Superior Técnico,*
10 *Universidade Técnica de Lisboa, 1049-001 Lisboa, Portugal*

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12

13 *correspondence author: palves@eq.uc.pt

14

14 Abstract

15 Prosthetic cardiac valves implantation is a common procedure used to treat heart
16 valve diseases. Although there are different prosthesis already available in the market
17 (either mechanical or bioprosthetic), their use presents several problems, specifically
18 concerning thrombogenicity and structural failure. Recently, some progresses have
19 been achieved in developing heart valves based on synthetic materials with special
20 emphasis in polymers. Among them, polyurethanes are one of the most commonly
21 used for the production of these devices.

22 Herein, Elastollan[®]1180A50, a thermoplastic polyurethane (TPU), was used to
23 formulate films whose surfaces were modified by grafting 2-
24 hydroxyethylmethacrylate (HEMA) either by Ultra-violet (UV) or by plasma
25 treatment. All films were analyzed before and after grafting. X-ray photoelectron
26 spectroscopy (XPS) measurements were used to evaluate TPU surfaces
27 functionalization. HEMA grafting was confirmed by the increase of the hydroxyl
28 (OH) groups' concentration at the surface of the films. Atomic force microscopy
29 (AFM) analysis was done to evaluate the surface topography of the biomaterials.
30 Results showed that the roughness of the surface decreased when HEMA was grafted,
31 especially for plasma treated samples.

32 After grafting the films' hydrophilicity was improved, as well as the polar component
33 of the surface energy, by 15 to 30%. Hydrophobic recovery studies using milli Q
34 water or PBS were also performed to characterize the stability of the modified
35 surface, showing that the films maintained their surface properties along time.
36 Furthermore, blood-contact tests were performed to evaluate haemolytic and
37 thrombogenic potential. The results obtained for HEMA grafted surfaces, using
38 plasma treatment, confirmed biomaterials biocompatibility and low thrombogenicity.

39 Finally, the cytotoxicity and antibacterial activity of the materials was assessed
40 through *in vitro* assays for both modified films. The obtained results showed
41 enhanced bactericidal activity, especially for the films modified with plasma.

42

43 Keywords: Heart valves, plasma and UV activation, polyurethanes, surface modification,
44 biocompatibility

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45 **1. Introduction**

46 Valvular heart diseases (VHD) include several heart conditions that can be either
47 congenital or acquired. Acquired VHD comprise degenerative valve diseases (which
48 are the most common in developed countries) and rheumatic heart sickness (mostly
49 common in developing nations) [1].

50 Nowadays, the number of patients diagnosed with degenerative valve disease is
51 progressively growing with population aging [2]. In fact, it is estimated that at least
52 one in each eight people over 75 years old will suffer from one kind of VHD,
53 becoming a serious public healthcare problem and a significant economic burden [3].

54 Some patients suffering from less severe valvular lesions are able to go through their
55 lifetime without ever needing surgical intervention. However, for others, surgery is
56 the only viable solution. Surgical treatment may involve the repair or the replacement
57 of the original damaged valve. The ideal choice would be to keep the original valve
58 [4]. However, for nearly 70% of the cases this procedure is no longer viable and valve
59 substitution must be performed [5]. Currently, nearly 280 000 heart valve substitutes
60 are implanted each year all over the world, in an approximated proportion of 50/50 for
61 mechanical and bioprosthetic valves [6]. Despite the improvements in the design and
62 composition of the commercially available valve prosthesis, mechanical valves have a
63 high associated risk of thrombogenicity, while the bioprosthetic valves may suffer
64 from premature structural failure [7]. Moreover, after a prosthetic valve implantation,
65 a life-threatening complication, known as prosthetic valve endocarditis (PVE), may
66 also occur [8]. PVE is usually caused by microorganism infection, especially bacteria
67 and fungi, and its treatment requires medical treatment. However, in advanced stages,
68 antibiotic therapy alone may not be enough to eliminate the infection and prosthesis
69 replacement is required [9].

70 In order to overcome these problems, synthetic leaflet heart valves have been
71 investigated over the last decades, trying to combine in one material, features like
72 durability and enhanced haemodynamics [10]. Among the different synthetic
73 materials used so far, polyurethanes (PU) have been considered the most suitable for
74 this purpose [11]. PU's properties such as abrasion resistance, affordable
75 manufacturing, chemical stability, durability, elasticity and haemocompatibility are
76 fundamental for their extended applications in the biomedical field [12]. Besides heart
77 valves, they are also used to prepare blood oxygenators, catheters, drug delivery
78 systems, internal lining of artificial hearts, scaffolds for tissue engineering and wound
79 dressing membranes [13]. However, several studies have reported that heart valves
80 produced with PU's, may suffer from premature failure, caused by their suboptimal
81 design and low durability [7]. The production of PUs with different compositions and
82 by applying different manufacturing techniques have resulted in materials with
83 improved properties which have allowed to expand their potential applications [14].
84 Thermoplastic polyurethanes (TPU) are currently being used for several industrial
85 uses, such as adhesives, coatings and films [15]. Furthermore, previous studies have
86 also shown their suitability for heart valves production [16].
87 Hereby, a commercial pre-processed polyether-based thermoplastic polyurethane
88 (Elastollan[®]1180A50), was studied in order to be applied in a near future as a base
89 material for synthetic heart valves manufacture. Elastollan[®]1180A50 choice was done
90 based on its absence of plasticizers, good heat resistance, high mechanical flexibility,
91 and its ability to be processed by moulding. Moreover, it also exhibits excellent
92 abrasion resistance, toughness, transparency, hydrolytic stability and fungal resistance
93 [17].
94 From the literature it is known that graft copolymerization of polyurethanes with

95 hydrophilic vinyl monomers, such as acrylic acid [18], acrylamide [19] and 2-
96 hydroxyethylacrylate [20], is an appropriate method to enhance surface
97 hydrophilicity, and improve its haemocompatibility [21].

98 Herein, 2-hydroxyethylmethacrylate (HEMA) was grafted onto the surface of
99 Elastollan[®]1180A50 to increase its hydrophilicity and improve its biological
100 properties. Plasma and UV irradiation were used to perform the modification of films'
101 surface. Furthermore, several parameters were assessed, such as their chemical
102 surface functionalities, roughness, antibacterial activity, blood compatibility,
103 cytotoxicity, hydrophilicity, hydrophobic recovery, surface energy and
104 thrombogenicity to evaluate their potential for being used in heart valve fabrication.

105

106

107 **2. Experimental**

108 **2.1. Materials**

109 Bacterial strain *Escherichia coli* (*E. coli*) DH5a was purchased from ATCC.
110 Elastollan[®]1180A50 was obtained from BASF. Fetal bovine serum (FBS) was
111 acquired from Biochrom AG (Berlin, Germany). Human Fibroblast Cells (Normal
112 Human Dermal Fibroblasts adult, criopreserved cells) were bought from PromoCell
113 (Labclinics, S.A.; Barcelona, Spain). LB agar was obtained from Pronadise.
114 Irgacure[®]2959 was kindly given by CIBA (Ciba Specialty Chemicals, Basel,
115 Switzerland). 3-(4,5-dimethylthiazol-2-yl)-5-(3carboxymethoxyphenyl)-2-(4-
116 sulphophenyl)-2H-tetrazolium, inner salt (MTS), amphotericin B, Dimethylformamide
117 (DMF), Dulbecco's modified Eagle's medium (DMEM-F12),
118 ethylenediaminetetraacetic acid (EDTA), 2-hydroxyethylmethacrylate (HEMA),
119 isopropyl alcohol, L-glutamine, penicillin G, phosphate-buffered saline solution

120 (PBS), streptomycin, and trypsin were acquired from Sigma-Aldrich (Sintra,
121 Portugal).

122

123

124 **2.2. Methods**

125 **2.2.1. Films preparation**

126 Elastollan[®] 1180A50 films were prepared by solvent evaporation. Elastollan[®] 1180A50
127 was dissolved in DMF to a 10% (w/v) TPU solution. This solution was poured into
128 glass Petri plates. Then, the Petri dishes were stored in an oven at 60 °C, for 24 hours.
129 Subsequently, films were removed from the dishes and ultrasonically cleaned with
130 isopropyl alcohol for 15 minutes, prior to surface grafting experiments.

131

132 **2.2.2. Argon plasma grafting**

133 A laboratory and small-scale production plasma system FEMTO (low pressure
134 plasma), manufactured by Diener Electronics, with a stainless steel plasma chamber
135 of 100 mm diameter and 270 mm length, was used for the plasma surface
136 modification experiments. TPU films were placed at 80 mm from the electrode and
137 were plasma treated with Argon, in a pressure chamber of 0.6 mbar, for 3 minutes and
138 applying 100 Watt of power to the electrodes to generate the plasma [22]. Then, the
139 plasma-treated TPU films were dipped into a 10% (v/v) aqueous solution of HEMA
140 and introduced in an oven at 60 °C, for 1 hour. Finally, the modified films (TPU-Ar-
141 HEMA) were washed abundantly with deionized water and dried until constant
142 weight was obtained.

143

144 **2.2.3. UV grafting with Irgacure[®]2959**

145 For the UV grafting, films were previously activated with UV light in a 0.5%
146 photoinitiator (Irgacure[®]2959) aqueous solution for 30 minutes. Afterwards, they
147 were removed and dipped into a 10% (v/v) HEMA aqueous solution. Then, samples
148 were irradiated with UV light during 30 minutes and the modified films were obtained
149 (TPU-UV-HEMA).

150 In both steps of the modification, films were irradiated using a Mineralight[®] Lamp,
151 Model UVGL-48, in the 254 nm wavelength setting. This generated a power of 6
152 Watt and the samples were placed at a distance of 4 cm from the light source.

153

154

155 **2.3. Characterization techniques**

156 **2.3.1. X-ray photoelectron spectroscopy**

157 X-ray photoelectron spectroscopy (XPS) measurements were made with a VGS
158 ESCALAB 200A spectrometer with an Al K α X-ray source. The operation conditions
159 were set to 15 kV. The binding energy scale was fixed by assigning a binding energy
160 of 285.0 eV to the -CH₂- carbon (1s) peak. The samples were analyzed at a take-off
161 angle of 0° relative to the normal of the surface. The C1s and O2s envelopes were
162 analyzed and peak-fitted using a combination of Gaussian and Lorentzian peak shapes
163 obtained from the XPS peak 4.1 software.

164

165

166

167

168 **2.3.2. Atomic force microscopy**

169 Atomic force microscopy (AFM) analysis of the samples was performed in a
170 Nanoscope IVa Veeco Metrology using the tapping mode (scan size 4.0 μm , scan rate
171 1.0 Hz). The average roughness (Ra) was calculated directly from the AFM images.

172

173 **2.3.3. Analysis of contact angle and surface free energy**

174 The contact angle and surface energy measurements were performed at room
175 temperature in an OCA 20 contact angle measurement unit from Dataphysics. Surface
176 free energy (γ_s) values as well as the dispersive (γ_s^D) and polar (γ_s^P) components
177 were obtained according to the Owens-Wendt-Rabel and Kaelbe method (OWRK) by
178 static contact angle measurements with three liquids: water, diiodomethane and
179 formamide. All measurements were performed on the air-facing surfaces of the films
180 with the three liquids using the sessile drop method. Nine measurements on different
181 points were performed on each sample from which the mean static contact angle and
182 its standard error were determined. The surface energies were assessed for all the
183 prepared films.

184

185 **2.3.4. Hydrophobicity recovery**

186 Samples were stored in vials containing milli Q water or PBS at 37 °C and examined
187 after 1, 2, 7, 15 and 30 days. Samples aged in air were wrapped in aluminium foil to
188 minimise hydrocarbon contamination, and those aged in milli Q water and PBS were
189 thoroughly washed with milli Q water and dried before analysis. The hydrophobicity
190 recovery was evaluated by water contact angle determination, as previously
191 described.

192 **2.3.5. Blood compatibility**

193 Blood compatibility assays were performed *in vitro* accordingly to the International
194 Standard Organization (ISO) 10993-4 [23]. Both the haemolytic potential and
195 thrombogenicity of the prepared films were evaluated.

196

197 *Haemolytic potential*

198 The haemolysis tests were performed as described in the American Society for
199 Testing and Materials (ASTM) F 756-00 standard [24]. Samples with 21 cm² were
200 placed in polypropylene test tubes and 7 mL of PBS (10 M, pH=7.4) were added.
201 After 72 h of incubation, at 37 °C, the PBS was removed and the samples were left to
202 dry. Then, 7 mL of PBS and 1 mL of diluted anticoagulated rabbit blood (ACD blood)
203 (10 mg/mL ± 1 mg/mL) was added to each sample. Positive and negative controls
204 were prepared by adding the same amount of ACD blood to 7 mL of water and PBS,
205 respectively. The tubes were placed at 37 °C, for 3 hours, and gently inverted twice
206 every 30 minutes to maintain materials in contact with blood. After incubation, the
207 fluid was transferred to a suitable tube and centrifuged at 700-800 g, for 15 minutes.
208 The amount of haemoglobin (Hb) released by haemolysis was determined by
209 measurement of the optical densities of the supernatants at 540 nm using a
210 spectrophotometer UV-vis (Jasco V550). The percentages of haemolysis (HI) were
211 calculated as described in equation 1.

$$212 \quad HI = \frac{[Hb]_{test} - [Hb]_{negativecontrol}}{[Hb]_{positivecontrol} - [Hb]_{negativecontrol}} \times 100 \quad \text{Equation 1}$$

213 According to the ASTM F 765-00 [24] materials can be classified as non-haemolytic
214 when $0 > HI > 2$, slightly haemolytic when $2 > HI > 5$ and haemolytic when $HI > 5$.

215

216

217 *Thrombogenicity*

218 The evaluation of thrombus formation on films surfaces (n=3 for each sample) was
 219 carried out using the gravimetric method of Imai and Nose [25]. Anticoagulated rabbit
 220 blood was also used for this purpose. Before performing the tests, the films were
 221 immersed in PBS solution (pH 7.4) at 37 °C. After 48 h of incubation, the PBS was
 222 removed and 250 µL of ACD blood were carefully placed over the surface of the
 223 films and also in an empty Petri dish, which acted as a positive control. Blood clotting
 224 tests were initiated by adding 25 µL of a 0.10 M calcium chloride solution and then
 225 stopped after 30 minutes, by adding 5 mL of water. The resultant clots were fixed
 226 with 1mL of a 36% formaldehyde solution and then dried with tissue paper and
 227 finally weighted. The percentage of thrombogenicity was determined by using
 228 equation 2.

$$229 \quad \% \text{ thrombogenicity} = \frac{m_{\text{test}} - m_{\text{negativecontrol}}}{m_{\text{positivecontrol}} - m_{\text{negativecontrol}}} \times 100 \quad \text{Equation 2}$$

230

231 **2.3.6. Evaluation of materials biocompatibility**

232 *Proliferation of human fibroblasts cells in the presence of the materials*

233 Human Fibroblasts cells were seeded in T-flasks of 25 cm² with 6 mL of DMEM-F12
 234 supplemented with heat-inactivated FBS (10% v/v) and 1% antibiotic/antimycotic
 235 solution. After cells attained confluence, they were subcultivated by a 3-5 minutes
 236 incubation in 0.18% trypsin (1:250) and 5 mM EDTA. Subsequently, cells were
 237 centrifuged, resuspended in culture medium and then seeded in T-flasks of 75 cm².
 238 Hereafter, cells were kept in culture at 37 °C in a 5% CO₂ humidified atmosphere,
 239 inside an incubator. To evaluate cell behavior in the presence of the materials,

240 fibroblasts cells were seeded with materials in 96-well plates at a density of 10×10^3
241 cells per well, for 96 hours. Previously to cell seeding, materials were firstly sterilized
242 using UV radiation for 30 minutes. Cell growth was monitored using an Olympus
243 CX41 inverted light microscope (Tokyo, Japan) equipped with an Olympus SP-500
244 UZ digital camera [26].

245

246 *Characterization of the cytotoxic profile of the films*

247 Human fibroblasts cells were seeded in the presence of materials, in 96-well plate,
248 with 100 μ l of DMEM-F12 and following incubated at 37 °C, in a 5% CO₂ humidified
249 atmosphere. After an incubation period (24, 48, 72 and 96 hours), cell viability was
250 assessed through the reduction of the MTS into a water-soluble formazan product.
251 Briefly, the medium of each well was removed and replaced with a mixture of 100 μ L
252 of fresh culture medium and 20 μ L of MTS/PMS reagent solution. Then, cells were
253 incubated for 4 hours at 37 °C, under a 5% CO₂ humidified atmosphere. The
254 absorbance was measured at 492 nm using a microplate reader (Sanofi, Diagnostics
255 Pauster). Wells containing cells in the culture medium without materials were used as
256 negative controls (K). Ethanol (96%) was added to wells that contained cells, as a
257 positive control (K⁺) [27].

258

259 **2.3.7. Characterization of the antibacterial activity of PU materials**

260 *Resazurin metabolic assay*

261 The resazurin assay was performed to evaluate bacterial growth in the presence of the
262 samples. Resazurin is a blue non-fluorescent and non-toxic dye that becomes pink and
263 fluorescent when reduced to resorufin by oxidoreductases within viable cells [28].
264 Briefly, bacterial cultures (*E.coli*) were grown overnight in culture medium without

265 antibiotics. The following day bacteria were seeded in 96 well plates at density of 5 x
266 10^6 colony-forming unit (CFU)/mL under aseptic conditions. Then, 10 μ L of 0.1%
267 resazurin solution were added to each well and the plate was incubated at 37 °C
268 during 1 to 4 hours. After, digital images of the plate were acquired to evaluate
269 bacterial growth using a Nikon digital camera (Nikon D50, Ayuthaya, Thailand).

270

271 **2.3.8. Statistical analysis**

272 The obtained results were expressed as the mean \pm the standard error of the mean
273 (n=4). Statistical significance was calculated using a one-way analysis of variance
274 (one-way ANOVA) and differences between groups were tested by a one-way
275 ANOVA with Dunnett's post hoc test [26].

276

277

278 **3. Results and Discussion**

279 **3.1. X-ray photoelectron spectroscopy**

280 To evaluate the existing functionalities on TPU surfaces (before and after the
281 grafting), wide scan and high-resolution XPS spectra were recorded. This analysis
282 was performed 24 hours after the grafting procedures. The elemental composition of
283 the surfaces was calculated from the XPS spectra.

284 As expected, the oxygen content of the grafted surface increased due to the grafting of
285 HEMA's hydroxyl groups (OH). Figure 1 shows the different spectra obtained for the
286 unmodified and the HEMA grafted TPU. Here, the C1s peak can also be resolved in
287 three components: the hydrocarbon (C-C and C-H) peak at 284.5eV, the ether peak
288 (C-O-C) at 285.9 eV and the urethane peak (NH-COO) at 288.0 eV. The ether peak

289 increases after the grafting and due to the addition of more ether peaks and to the
290 addition of OH groups present in the HEMA monomer structure. The increase of this
291 peak is slightly higher when the UV grafting method was used. All these changes
292 suggest that the surfaces were successfully grafted. The same conclusion can be
293 inferred by looking at the O1s peak, which is resolved in two peaks: the C=O peak at
294 532 eV (which was already present in the untreated TPU) and the C-OH peak at 534
295 eV due to the hydroxyl groups [29]. From the differences observed in the peak areas
296 and the relative composition ratio based on the area of each peak presented in Figure
297 1 can be easily concluded that the plasma grafting method is more efficient than the
298 UV. Also, it can be implied that the graft density is higher for the plasma treated
299 surface since more OH groups are present on the surface as a consequence of the
300 grafting of HEMA.

301

302

303 **3.2. Atomic force microscopy (AFM)**

304 Materials surface topography was evaluated by AFM analysis. Figure 2 shows the 3-
305 dimensional AFM images of unmodified TPU and grafted TPU.

306 It is known that a surface becomes smoother after grafting, when compared with its
307 original state, depending on monomer bounded to the surface [30]. Therefore, as
308 expected from the results obtained in XPS analysis, distinct surface topographies were
309 observed depending on the employed grafting method. The micrographs from Figure
310 2 show that when plasma is used, the resulting surface is smoother than when using
311 UV grafting treatment, which indicates that a higher graft density was obtained when
312 plasma was used for the grafting reaction. The higher graft density results in a

313 smoother surface. This difference between both grafting methods has already been
314 observed for the grafting of other monomers in previous works [31].

315 The roughness of the materials surfaces was determined to better quantify the
316 differences between them. The average roughness (Ra) was calculated directly from
317 the AFM images in a 700×700 nm surface region. The obtained Ra results, presented
318 in Figure 2, confirm that the roughness of the surface decreases when HEMA is
319 grafted, especially when the plasma method is used.

320

321

322 **3.3. Water contact angle and surface energy measurements**

323 It is widely recognised that surface energy is an important parameter affecting
324 polymers adhesion, material wettability and even biocompatibility [32]. The
325 measurement of contact angles is considered as the most convenient method for
326 determining the surface free energy of solid samples. This technique relies on the
327 determination of the interactions between the solid sample of interest and liquids with
328 well determined surface tensions.

329 According to Owens, Wendt, Rabel and Kaelble the interfacial tension can be divided
330 in two components: dispersive interactions and polar interactions [33]. Polar
331 interactions comprise Coulomb interactions between permanent dipoles and the ones
332 between permanent and induced dipoles. The interactions caused by time fluctuations
333 of the charge distribution within the molecules are called dispersive interactions.

334 Table 1 summarizes the obtained results for water contact angles and surface energies
335 as well as the percentages of the polar components of the surface energies for the
336 unmodified and modified TPU films.

337 The presence of polar functional groups such as OH or NH₂ increases the hydrogen
338 bounding interactions [34] and therefore decreases water contact angle and increases
339 the polar component of the surface energy. For this reason, in this study, a decrease in
340 the water contact angle could be observed after HEMA grafting. The water contact
341 angle of the surface decreases from 82.7° for the unmodified surface, to 75.0° and
342 60.8° after grafting HEMA by UV and plasma, respectively. Such results are ascribed
343 to the grafting of HEMA's OH groups, and this variation is even higher when the
344 plasma grafting method is used. Moreover, the obtained results for the surface energy
345 presented in Table 1 show an increase in the polar component of the surface energy
346 when films surfaces are grafted with HEMA. While the original TPU film presents a
347 polar component of 7.1%, this value changed to 15.5% and 38.7% when HEMA was
348 grafted onto the surface by UV irradiation and plasma treatment, respectively. The
349 results obtained for films treated with plasma suggest that the efficiency of grafting is
350 higher when this method is used. Furthermore, these results are in accordance with
351 those obtained by XPS, AFM and were further validated by the thrombogenicity
352 studies.

353

354

355 **3.4. Hydrophobic recovery analysis**

356 Hydrophobic recovery was evaluated by static water contact angle measurements for
357 a period of 30 days. The water contact angles were determined before and after each
358 modification and along time by incubating samples in different mediums (air, milli Q
359 water and PBS). The obtained results are presented in Figure 3. TPU has a water
360 contact angle of $82.7^\circ \pm 1.8$; after the grafting procedure this value decreases to 75.0°
361 ± 0.4 and $60.8^\circ \pm 1.7$ when UV and plasma methods were used, respectively. Plasma

362 method led to lower values, suggesting once again, that this method is more efficient
363 for grafting HEMA than the UV method. The decrease in the water contact angle is
364 ascribed to the HEMA polar groups (OH groups) present on TPU surface after
365 grafting. These groups increase dipole–dipole interactions and therefore increase
366 hydrophilicity. For this reason, the observed decrease in the water contact angles may
367 represent an indirect measure of the extent of modification.

368 From the hydrophobicity recovery profiles shown in Figure 3, it can be seen that after
369 the grafting reactions, despite the storage medium, surfaces partially recovered their
370 hydrophobicity along time. The hydrophobicity recovery of a surface might be
371 explained by air contamination or even by surface rearrangements [30,35,36].
372 Comparing both grafting methods, this recovery is more evident for surfaces grafted
373 by the UV method, suggesting that these surface modifications are not stable. When
374 comparing storage mediums, the air is more prone to allow the hydrophobicity
375 recovery, while milli Q water or PBS maintain the surface properties, meaning that
376 storing the grafted surfaces in PBS or milli Q water prevents surface rearrangements
377 and mainly eliminates air contaminations preventing the hydrophobicity recovery.

378

379

380 **3.5. Blood compatibility**

381

382 **3.5.1. Haemolytic potential**

383 Haemolysis is regarded as an especially significant screening test, once it provides
384 quantification of small levels of plasma haemoglobin that may not be measurable
385 under *in vivo* conditions [37].

386 The haemolysis index (HI) of the original and modified TPU films was determined

387 according to ASTM F 756-00 [24], by using the cyanmethemoglobin method to
388 quantify the haemoglobin (Hb) present in the supernatant after the films were
389 incubated with blood.

390 Based on the results obtained, both the unmodified and the modified films are
391 classified as non-haemolytic. In fact, HI is very similar for all samples varying from
392 1.85 ± 0.11 (for the original film) to 1.88 ± 0.05 for TPU-UV-HEMA and finally to
393 1.36 ± 0.51 for TPU-Ar-HEMA. This means that surface modification with the HEMA
394 molecule does not induce any damage in erythrocytes' membranes that could lead to
395 their lysis. Although some literature states that it is not possible to define a universal
396 level of acceptable or unacceptable haemolysis values [29], by definition, a blood-
397 compatible material should not promote haemolysis. In this work this parameter is of
398 extreme importance since the material will be contacting directly with blood during
399 its entire lifetime. It is therefore safe to say that the modified films will not be
400 responsible for the lysis of the erythrocytes once implanted in the human body. Such
401 feature is fundamental for the desired biomedical application.

402

403

404 **3.5.2. Thrombogenicity**

405 Serum proteins adsorption onto the surface of a material is a key phenomenon for the
406 thrombogenic process. In fact, thrombus formation is inversely proportional to blood
407 compatibility of a given material. This parameter is of extreme importance when
408 materials are designed to be used in direct contact with the blood stream as it is the
409 case of heart valves.

410 The induction of thrombus formation on the surface of the prepared films was
411 evaluated by gravimetry after 30 and 60 minutes of contact with blood. Both the

412 unmodified and modified TPU films (TPU-UV-HEMA and TPU-Ar-HEMA) were
413 tested using three samples of each set. The resultant weights of the blood clots from
414 these tests were obtained and the percentage of thrombogenicity was calculated. In
415 Figure 4 it is possible to verify that the surface modification methods used influence
416 the thrombogenicity of the materials. As depicted in this Figure, both unmodified
417 TPU as well as TPU-UV-HEMA films are highly thrombogenic after 30 and 60
418 minutes of being in contact with blood. On the other hand, the TPU-Ar-HEMA films
419 presented a much lower value for thrombus formation for both incubation times
420 (19.33 and 21.2% after 30 and 60 minutes, respectively). These results are coherent
421 with those obtained for the surface energies values. It has been stated that
422 thrombogenicity is directly related with the value of surface energy presented by a
423 surface [38]. Interestingly, in this case, the surface energies of the films are not
424 significantly different between them. However, the polar component of surface
425 energy changes considerably. As previously described, it was verified that the polar
426 component of the surface energy was higher for the TPU-Ar-HEMA film, suggesting
427 that larger amounts of hydrophilic HEMA molecules were grafted on the TPU film,
428 when plasma method was used. Hydrophilic surfaces are usually associated to low
429 proteins adsorption since they adsorb weakly and reversibly to these types of surfaces
430 [38]. Considering that protein adhesion constitutes the first step of coagulation
431 cascade that ends in thrombus formation [39], it can explain the distinct thrombogenic
432 character presented by films studied here.

433

434

435 **3.6. Characterization of materials biocompatibility**

436 Films cytocompatibility was evaluated through *in vitro* studies. Human fibroblasts

437 cells with the same initial density were seeded in the 96-well plates, with or without
438 materials to assess films cytotoxicity. Cell adhesion and proliferation in the presence
439 of the materials was characterized through optical microscopy by using an inverted
440 microscope. Figure 5A shows that cells adhered and proliferated in the presence of all
441 the materials and also in the negative control. Such results demonstrate that all films
442 are biocompatible.

443 In order to characterize the physiological response of cells to the presence of the
444 TPUs, a MTS assay was also performed. The MTS assay results (Figure 5B) showed
445 that cells maintained a similar viability to the ones cultured in the absence of films
446 during 96 hours. Furthermore, cells presented higher viabilities in the presence of
447 modified TPUs during the first 72 hours.

448 The results obtained herein show that the modifications performed on the TPU
449 surface did not affect cell integrity or viability, a fact that is crucial for the proposed
450 biomedical application, *i.e.*, to be used as heart valves.

451

452

453 **3.7. Antibacterial activity**

454

455 **3.7.1. Resazurin assay**

456 The antimicrobial activity of TPUs was evaluated through a resazurin reduction assay.

457 As demonstrated in Figure 6, TPU-Ar-HEMA presented bactericidal activity, whereas

458 unmodified TPU and TPU-UV-HEMA did not show any significant activity. Such

459 results are of crucial importance, because it seems that HEMA grafted TPU by plasma

460 treatment may contribute to avoid the biofilm deposition on this material once

461 implanted inside the human body and therefore avoid common and dangerous
462 complications such as prosthetic valve endocarditis.

463

464 **4. Conclusion**

465 The grafting of HEMA onto the surface of Elastollan[®] 1180A50 films (TPU) by UV
466 irradiation and Argon low pressure plasma treatment led to a modification their
467 properties. This modification improved some of the films properties, and the resulting
468 materials are good candidates to be used for heart valves production. Additionally,
469 Argon plasma treatment showed to be more efficient for the grafting of HEMA when
470 compared to the UV method. The TPU-Ar-HEMA surface showed higher content of
471 OH groups indicating a higher HEMA density, which led to a smoother and more
472 hydrophilic surface. Also, Argon plasma treatment showed significantly lower values
473 of thrombogenicity in comparison with those of unmodified and UV modified TPUs.
474 Furthermore, the modified films upheld their intrinsic biocompatibility and, more
475 importantly, enhanced the bactericidal activity of the materials. This fact was once
476 again more evident for TPU-Ar-HEMA.

477 Based on the overall results it may be concluded that TPU-Ar-HEMA is a good
478 candidate to be used in a near future for the preparation of prosthetic heart valves.

479

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485

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574

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576

577

577 **List of Tables**

578 Table 1: Water contact angle (θ), surface energy (γ_s), dispersive (γ_s^D) and polar
579 components (γ_s^P) of the surface energy and % of polar component of the unmodified
580 and modified TPU films. Each result is the mean \pm standard error of the mean of three
581 independent experiments.
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582 **Figures captions**

583

584 Figure 1: XPS spectra of C1s (a) O1s (b) for TPU (1) and the TPU-Ar-HEMA (2) and
585 TPU-UV-HEMA (3). The relative composition ratio based on the area of each peak
586 for the all TPU films are also indicated.

587

588 Figure 2: Atomic force micrographs of the TPU and HEMA grafted surfaces. Surface
589 Roughness, Ra (nm) is indicated as the mean \pm standard error of the mean of three
590 independent experiments.

591

592 Figure 3: Hydrophobic recovery of TPU, TPU-Ar-HEMA and TPU-UV-HEMA
593 stored in air, milli Q water and PBS. All samples were stored in the mediums after
594 grafting. Each result is the mean \pm standard error of the mean of three independent
595 experiments.

596

597 Figure 4: Values of thrombus mass obtained for original and modified TPU films after
598 30 and 60 minutes of blood contact. Each result is the mean \pm standard error of the
599 mean of three independent experiments.

600

601 Figure 5: A) Microscopic photographs of human fibroblasts cells seeded in the
602 presence of the different PU materials (*) for 24, 48, 72 and 96 hours of incubation,
603 original magnification x100. B) Evaluation of the cellular activity after 24, 48, 72 and
604 96 hours in contact in TPU materials. Positive control (K^+); negative control (K^-).
605 Each result is the mean \pm standard error of the mean of three independent
606 experiments. Statistical analysis was performed using one-way ANOVA with
607 Dunnet's post hoc test (* $p < 0.001$).

608

609 Figure 6: Evaluation of the TPU-UV-HEMA (upper line), TPU-Ar-HEMA (middle
610 line) and TPU (lower line) films anti-bacterial effect. Image represents the
611 colorimetric result of a resazurin assay, performed after culturing *E. coli* (DH5 α) with
612 the different materials for 24 hours. For each material, several quantities were tested
613 and bacterial culture controls were also performed, which are represented by K⁺ for
614 the negative control (blue, represents dead bacteria) and K⁻ for the positive control
615 (pink, represents viable bacteria).

616

616

617 Table 1

	θ (°)	γ_s (mN/m)	γ_s^D (mN/m)	γ_s^P (mN/m)	% of γ_s^P
TPU	82.7 ± 1.8	35.99 ± 2.44	33.43 ± 2.33	2.56 ± 0.71	7.1
TPU-UV-HEMA	75.0 ± 0.4	36.55 ± 1.51	30.70 ± 1.27	5.65 ± 0.81	15.5
TPU-Ar-HEMA	60.8 ± 1.7	38.31 ± 2.50	23.53 ± 1.68	14.78 ± 1.86	38.7

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620 **Highlights**

621 • PU films were modified grafting HEMA by UV irradiation and Ar low
622 pressure plasma

623 • This novel functionalization of PUs improved their blood compatibility

624 • Modification with plasma significantly lowered values of thrombogenicity

625 • The modified films upheld their intrinsic biocompatibility

626 • Surface treatments enhanced the bactericidal activity of the materials

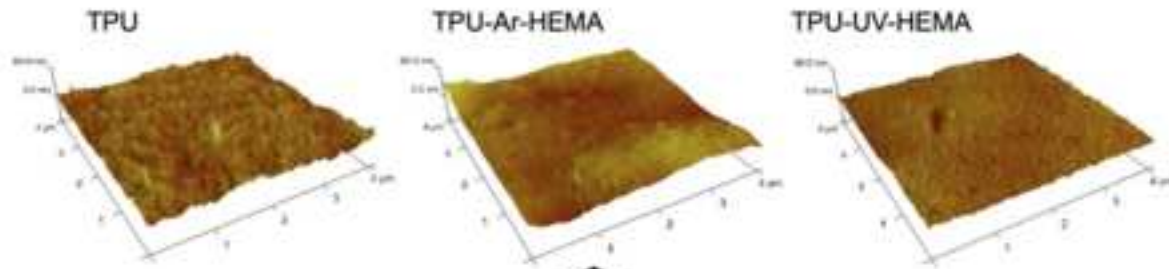
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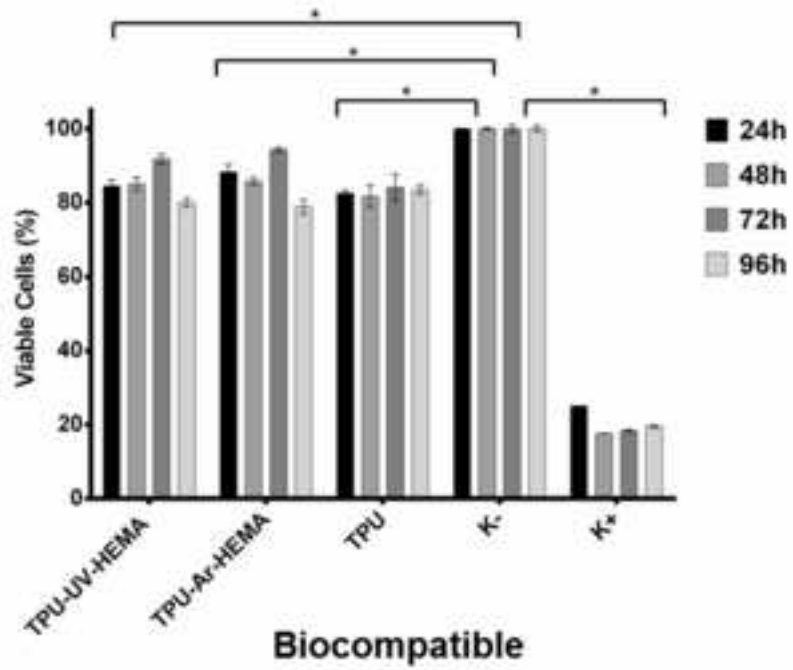


Thermoplastic urethane (TPU) film

↓ HEMA grafting

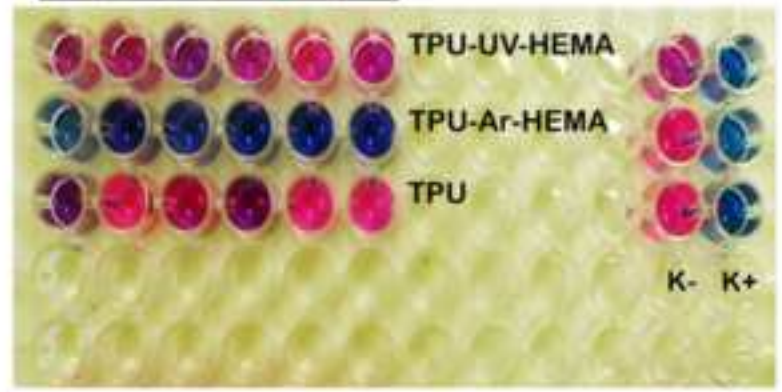


MTS assay



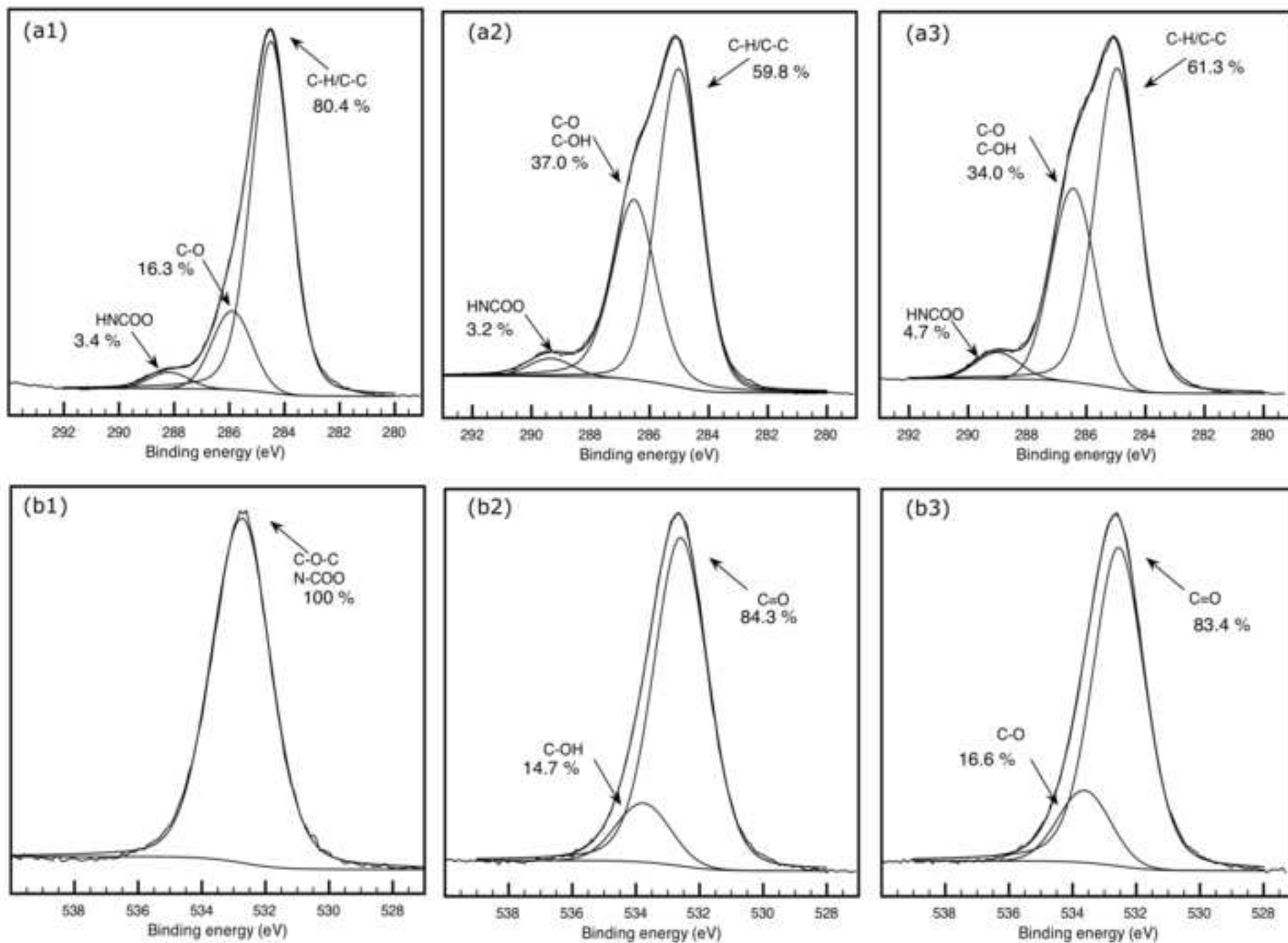
Biocompatible

Biomaterial Weight (mg)
20 15 10 8 6 3

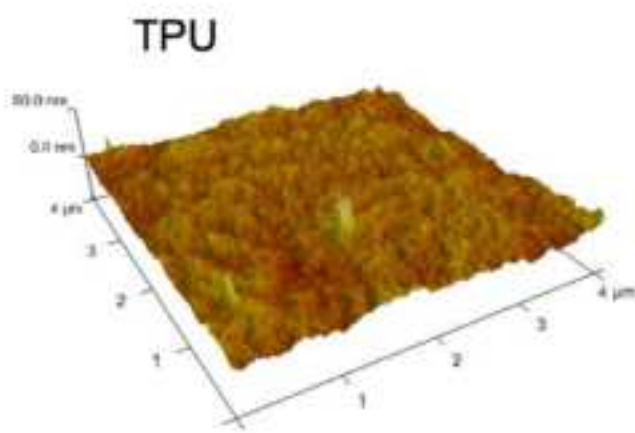


Bactericidal effect

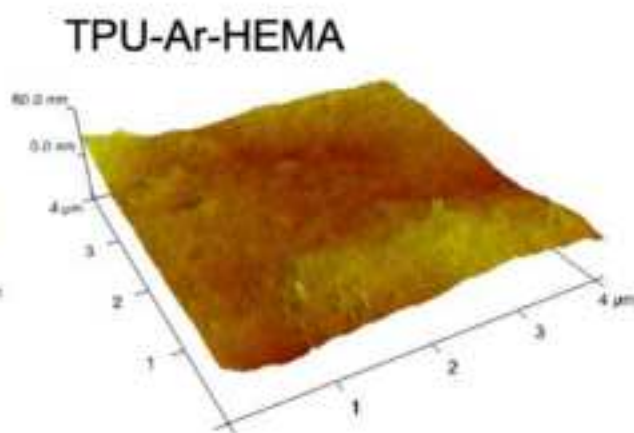
Figure 1



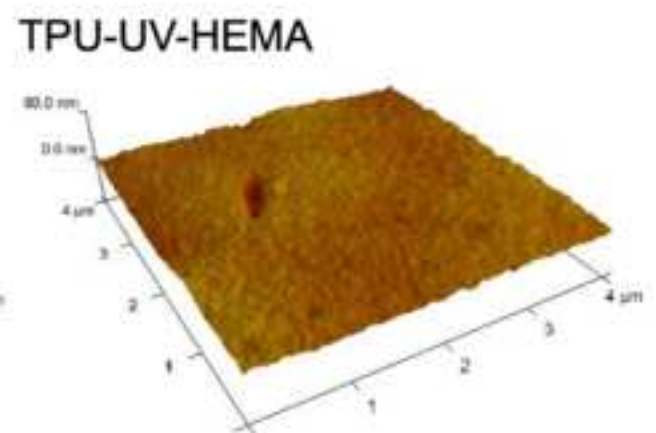
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$Ra = 2.57 \pm 0.31$



$Ra = 1.70 \pm 0.19$



$Ra = 1.82 \pm 0.13$

Figure 3

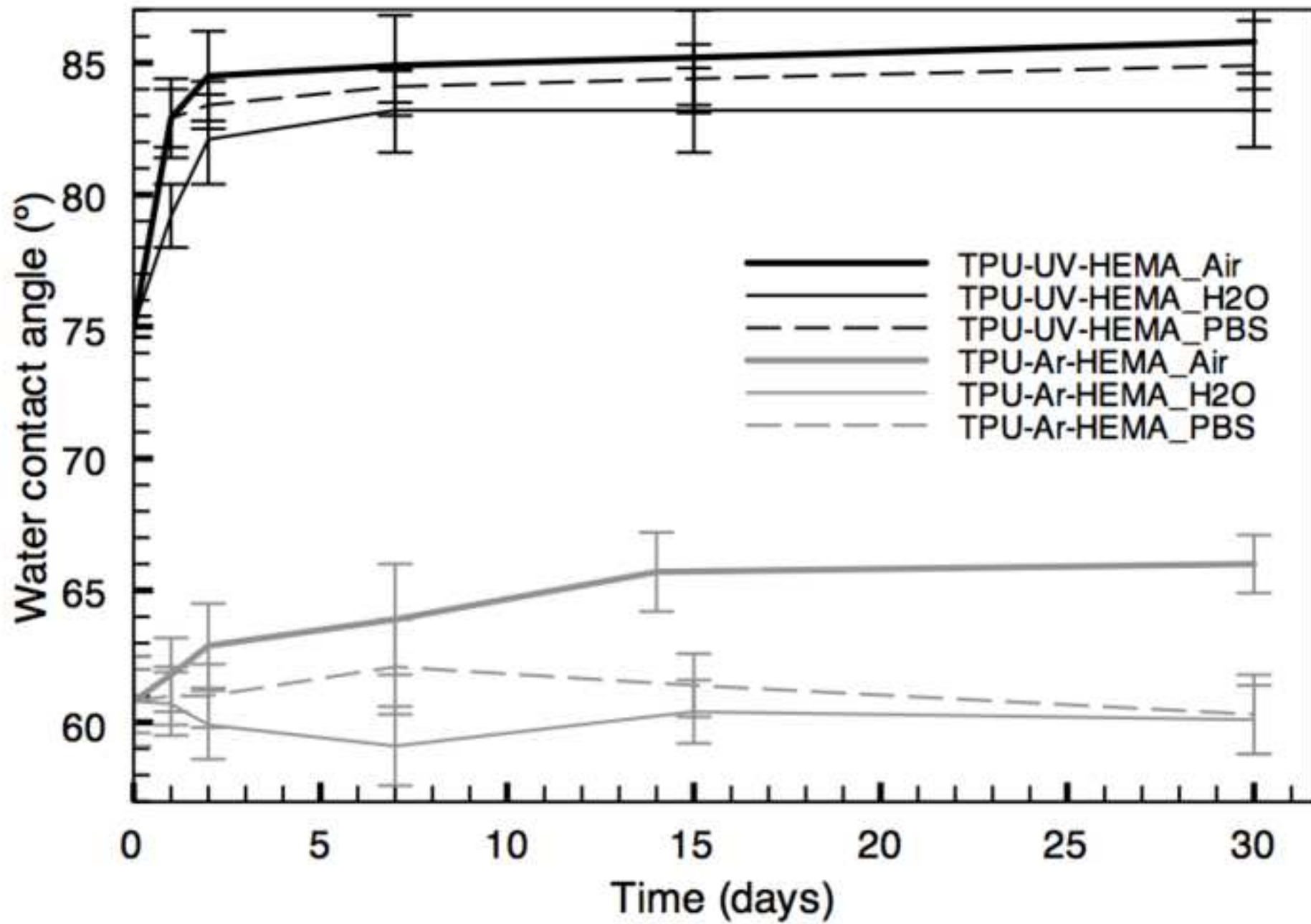
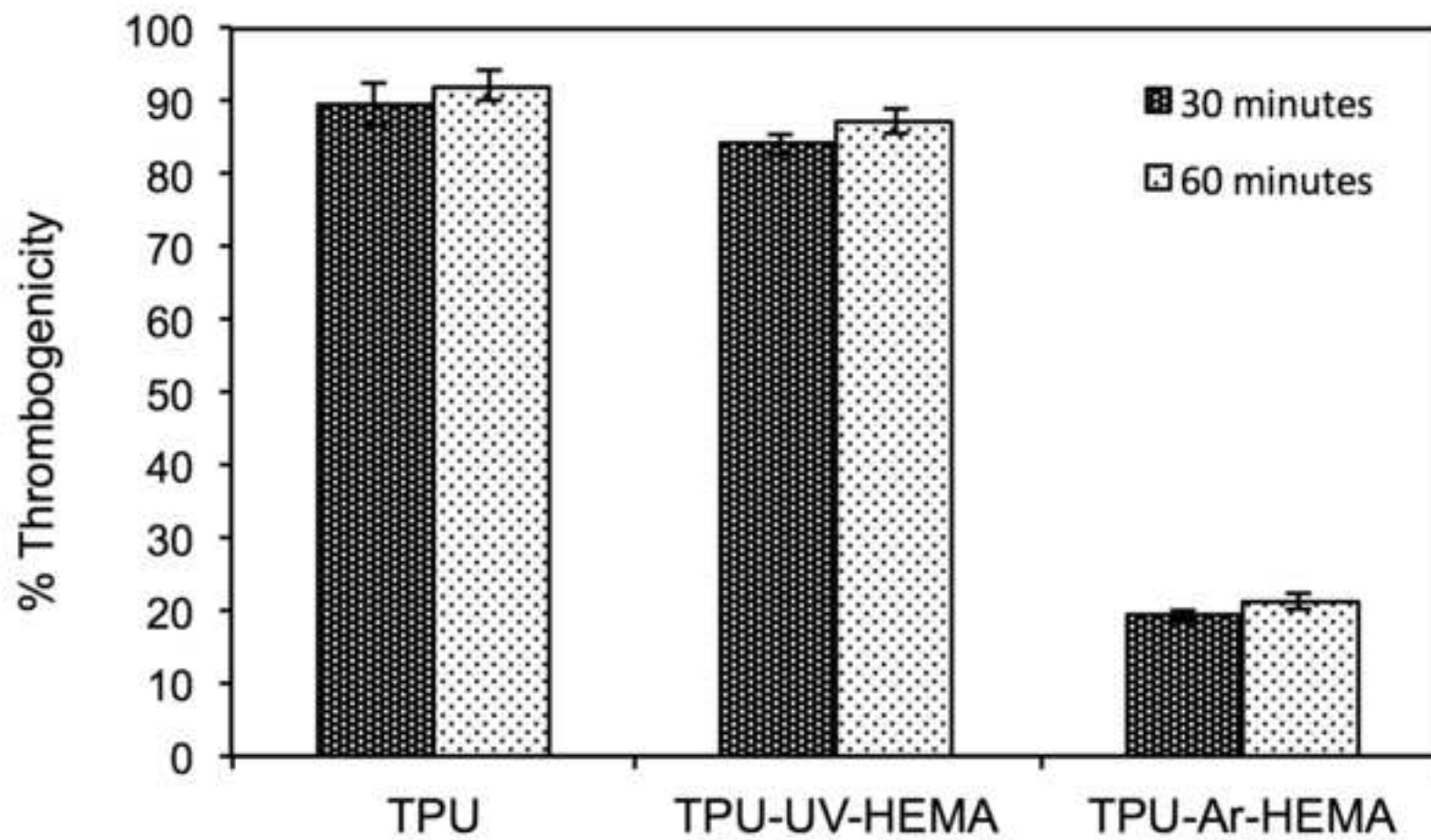


Figure 4



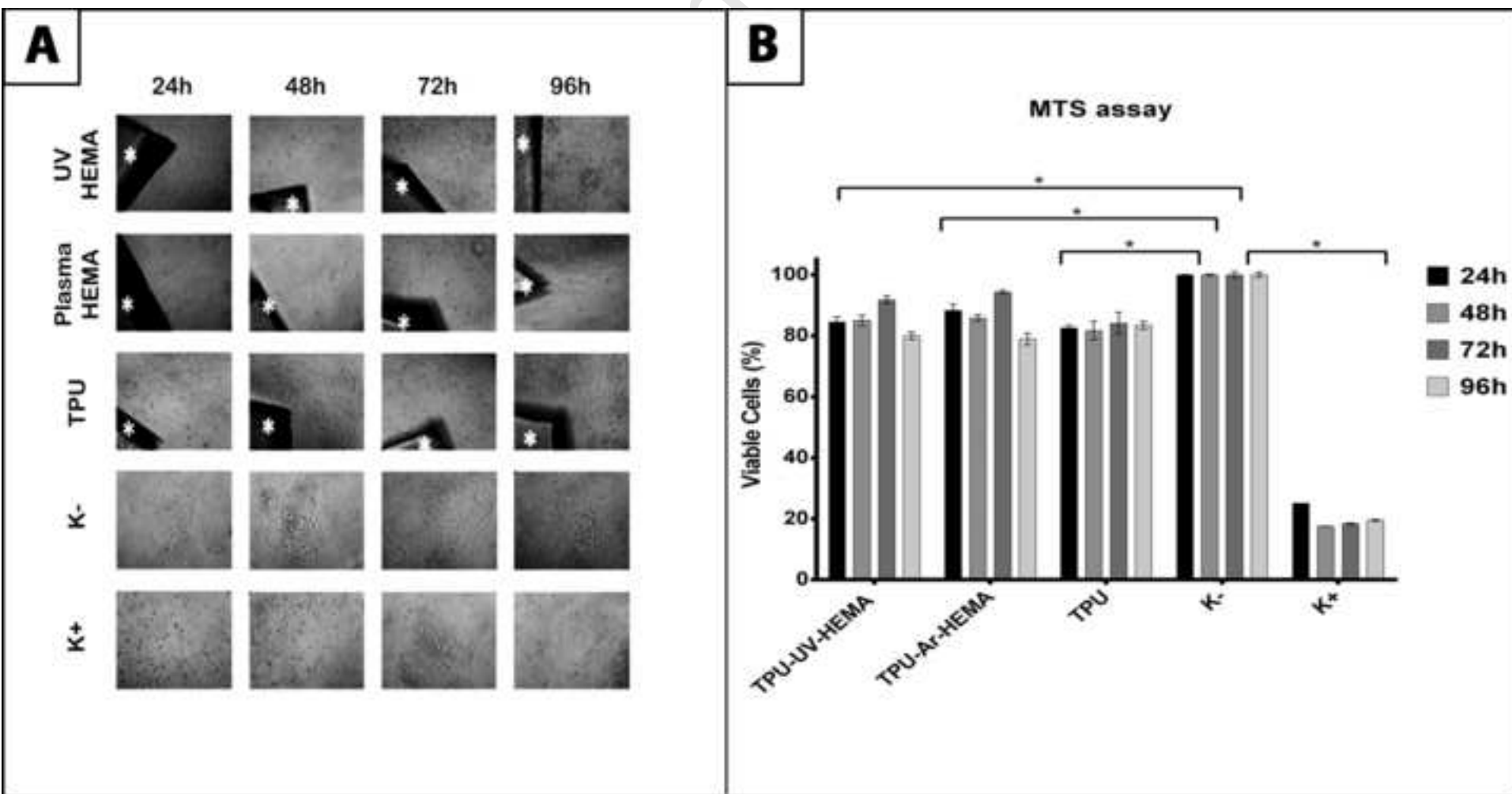


Figure 6

Biomaterial Weight (mg)

20	15	10	8	6	3
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